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Keywords:	X-ray computed tomography, Additive Manufacturing, 3D printing, Composites, Multi-material

X-ray computed tomography evaluations of additive manufactured multi-material composites

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Abstract

Additive Manufacturing (AM) often produces complex engineered structures by precisely distributing materials in a layer by layer fashion. Multi-material AM is a particularly flexible technique able to combine a range of hard and soft materials to produce designed composites. Critically, the design of AM multi-material structures requires the development of precise three-dimensional (3D) computer aided design (CAD) files. While such digital design is highly employed, techniques able to validate the physically manufactured composite against the digital design from which it is generated are lacking for AM, especially as any evaluations must be able to distinguish material variation across the 3D space. In this paper, a non-destructive approach using X-ray computed tomography (XCT) is used to fully evaluate the 3D distribution of multi-materials from an AM process. Specifically, two diverse hard and soft materials are alternatively produced in the form of a fibre embedded in a matrix. XCT coupled with imaging evaluation were able to distinguish between the differing materials and, importantly, to demonstrate a reduction in the expected fabricated volumes when compared to the respective CAD designs.

Keywords: X-ray computed tomography, Additive Manufacturing, 3D printing, Composites, Multi-material.

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34 **Introduction**

35 Three-dimensional (3D) additive layer manufacturing (AM), commonly referred to as
36 3D printing, has shown significant potential in fabricating complex parts. The attractiveness of
37 AM is based on digital designs, with almost arbitrary shape and complexity being translated to
38 a physical model through additive material deposition¹. Limits in free form fabrication of
39 classical manufacturing such as mould jetting and CNC machining have been overcome using
40 AM approaches^{2 3 4 5}. The flexibility of AM lies in the development of computed aided design
41 (CAD) that are subsequently translated in standard tessellation language (STL), which controls
42 the physical production of either simple mono-material or more organised multi-material parts.
43 For the latter, multi-material AM has attracted the attention of material scientists due to its
44 ability to construct engineered composites. The recent availability of multi-jetting technology
45 is of significant importance and, as a result of this importance, has led to the recognition within
46 ISO and ASTM international standards^{6 7 8}. However, techniques able to validate the digital
47 designs of the AM composite with the ability to evaluate compositional changes across the 3D
48 space are lacking. Common inspection methods, such as optical microscopy (OM) and
49 scanning electron microscopy (SEM), are widely used to provide detailed topological and
50 compositional information inspecting the surface of a sample before or after testing.
51 Compositional information can be further provided integrating SEM with an energy Dispersive
52 X-ray spectrometry systems (EDS)⁹ but both SEM and EDS fundamentally allow two-
53 dimensional (2D) inspections only.

54 The work herein presented attempts to define a suitable method of evaluating AM
55 multi-material composites. While the approach is flexible for a range of composite structures,
56 consideration of an AM material jetting system is provided. Material jetting systems exploit a
57 piezoelectric ink-jet print head to precisely deposit on demand nano-droplets of photo-curable
58 materials containing a photo initiator. A photo-polymerization system is coupled to the
59 piezoelectric printing head to cure the droplets which then form a solid polymer, layer by layer,
60 onto a building platform known as substrate^{10 11}. The light coming from an ultraviolet (UV)
61 lamp activates the curing process of the photopolymer layers, each one sticking to the previous
62 layer until the entire designed part is realised. Typically, these materials are similar in physical
63 characteristics such as density, which makes attempts to differentiate them a challenge using
64 OM and SEM unless variations in surface finish or colour are apparent.

65 In previous studies¹² optical and SEM investigations were useful to monitor mechanical
66 testing of multi-material composites, which allowed computational predictions on crack
67 development and related energy dissipation/toughening strategies of biologically inspired

68 structures^{13–14}. In another study, two materials of contrasting hard and soft mechanical
69 properties were combined to explore the effects of simple 3D chess patterns, printed in a voxel-
70 based manner with varying dimensional parameters to influence the fracture path. SEM was
71 employed to routinely highlight contours of different regions as well as failure lines
72 predominantly propagating through the rigid material during mechanical testing¹⁵.

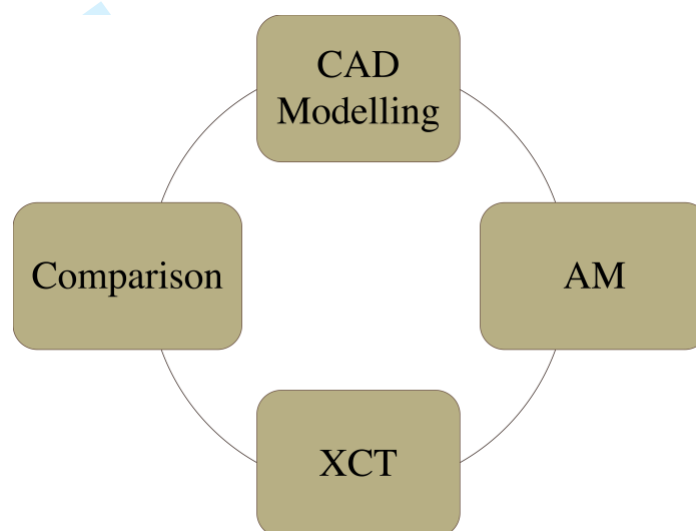
73 Despite imaging techniques such as OM and SEM are able to provide topological
74 information on the surface of shaped materials, complementary imaging techniques such as X-
75 ray computed tomography (XCT) are necessary to reveal volumetric information where the
76 other imaging techniques are not practical. Probed samples, whether they are single objects,
77 hierarchical assemblies of materials, exhibit internal features that are detected and
78 reconstructed using computed image techniques based on the interaction of X-rays with matter.
79 Indeed, XCT generates 3-dimensional representations of objects, typically non-destructively,
80 hence allowing sample monitoring before, during and after a particular testing regime is applied
81 to the sample. As composites are used predominantly in structural applications, mechanical
82 testing is usually applied to evaluate deformation and failure of the various constituent
83 materials as well as the resultant structure^{16–19}. XCT allows mapping of the effective density
84 distribution in the fabricated part, which predominantly defines the attenuation between the
85 probe X-ray and material. The generated 3D map consists of voxels that are displayed with a
86 greyscale value related to the X-ray attenuation of the material^{20–21}. If materials identification
87 is possible, CAD designed features and triangulated surfaces (meshes) resulting from the image
88 processing after XCT scanning can be compared. So far, the use of XCT combined to AM has
89 been primarily oriented to investigate and compare mono-material AM parts against CAD
90 models, thus considering topology primarily. Routine studies have evaluated the porosity of a
91 structure produced from a stereolithography AM process using XCT imaging where the
92 material and porous space provided significant variation in the X-ray probe attenuation with
93 the sample²². XCT investigation of AM multi-material structures have been performed to
94 analyse the location, for example, of active pharmaceutical ingredients compacted within a 3D
95 printed carrier²⁰, evaluate the interfacial density variation between copper and steel and
96 examine alloying of molybdenum particles in titanium parts produced via laser powder bed
97 fusion²³. In some cases, synchrotron XCT techniques were used to accurately determine
98 distribution of pores and inclusions in the same sample manufactured with laser metal
99 deposition²⁴.

100 Currently the use of XCT to determine composition of AM multi-material structures
101 has been neglected especially when employing multi-jetting technology to facilitate the

102 production of polymeric composite structures where different materials within such structures
103 have similar densities^{25 26 27}. Understanding the variation in composition for AM multi-material
104 structures has been shown to be important in determining the fracture development of the
105 structure during mechanical loading. Composite materials incorporating bone, biocalcite and
106 bone-like geometries^{28 29 30 31} were produced using ink-jetting AM and avoiding multistep parts
107 assembly of soft phase and hard-stiff inclusions to reproduce the mechanical interplay between
108 phases and toughening mechanisms inspired by nature.

109 In the work presented here, XCT is employed to qualitatively and quantitatively analyse
110 the assembly of ink-jet multi-material AM composites made of soft and hard phases.

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Figure 1. XCT-based workflow to approach and inspect AM composite parts.

114

115 A composite structure comprising of a single hard fibre feature embed in a softer matrix
116 material was imaged using XCT. Reconstructed images were analysed in terms of surface area
117 and volume to be compared to the original CAD sources according to the schematic in Figure
118 1. A critical aim of the evaluations is the potential for XCT to distinguish between different
119 UV curable 3D printing materials but similar in density, hence in X-ray attenuation. Hard and
120 soft materials were allocated to both matrix and inclusion to detect any difference in contrast
121 and shape. The proposed XCT-based workflow aims to compare input drawings with the
122 manufactured ones, maximizing contrast between the fibre and matrix.

123

124 **Material and Methods**

125 **Design and 3D printing**

126 An interlocking single fibre composite was designed in Rhinoceros 3D 5.0 (Robert McNeel
127 and Associates, USA) in the form of complementary regions defined by precise geometrical
128 boundaries. The interlocking shape was chosen to allow warping of thin walls as a result of the
129 manufacturing process, which will then be highlighted within the geometrical difference
130 between the CAD design and the XCT reconstruction. The resultant CAD model is displayed
131 in Figure 2a, where a rectangular feature is included in a matrix characterised by narrow neck
132 (2 mm) and a wide body (10 mm). The nominal thicknesses of the matrix and fibre are 3 mm
133 and 2 mm, respectively. An inkjet-based 3D printer (ProJet 5500X, 3D Systems, USA) allowed
134 for the simultaneous layered deposition of two base materials, namely a hard-white material
135 (VisiJet® CR-WT 200) and a soft black material (VisiJet® CE-BK) from the same 3D printer
136 manufacturer. These materials are referred to in this paper as WT and BK, respectively. The
137 3D printer resolution was set to an ultra-high definition 13 µm layer thickness (750 x 750 x
138 2000 Dots per Inch (DPI)), which is the highest performance of the 3D printer used in this
139 study. Two sample configurations of black fibres in a white matrix (BK into WT) and white
140 fibres in a black matrix (WT into BK) were designed (Figure 2b).

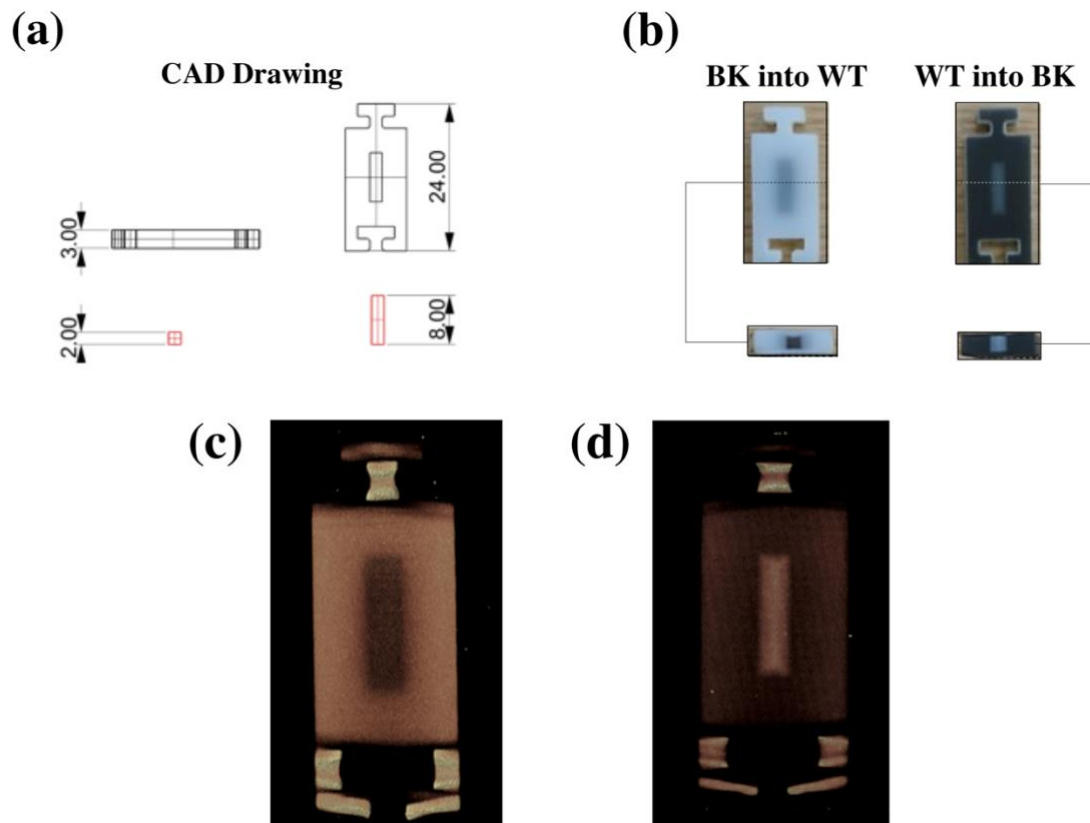
141 **XCT evaluation**

142 Evaluations of the 3D printed samples were carried out using an XCT system (Versa 520, Carl
143 Zeiss Microscopy Ltd., USA) operating with a 70 kV/6 W. Polyurethane foam material was
144 used to support and hold in place the parts within a box confining the volume of interest within
145 the XCT system. A 29 µm isotropic voxel size was achieved from imaging samples with a total
146 of 1601 projections over 360°. Each projection was collected at a 2 s exposure time. The
147 reconstruction of image datasets of the two scanned samples was performed using Scout and
148 Scan software (Zeiss) and rendered using XM3DViewer 1.2.8 (Zeiss). Figures 2c and 2d depict
149 the output of the rendering showing the difference between a BK into WT and WT into BK
150 assembly.

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154

155 Figure 2. Representation of the interlocking fibre composite. a) CAD drawing of the multi-material part. The
 156 matrix is reported in black, the inner fibre in red. All the dimensions are in mm. b) Fabricated parts with
 157 complementary materials allocation. XCT reconstruction of the corresponding composites (c) BK in WT
 158 and (d) WT in BK.

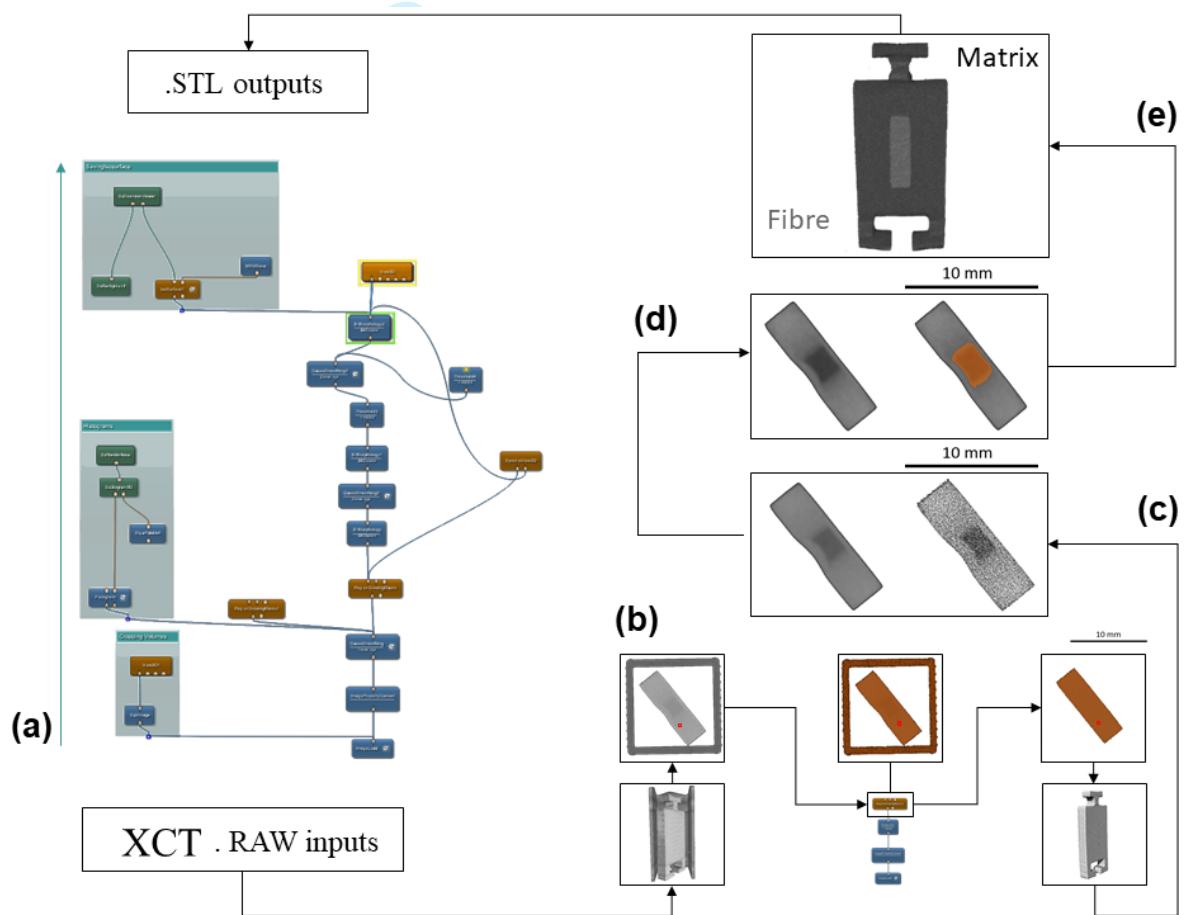
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160 Image post-processing

161 The resultant XCT images were post-processed using a semi-commercial software (MeVisLab,
 162 MeVis Medical Solutions AG, <http://www.mevislab.de/>). The post-processing was applied to
 163 enhance contrast between the WT and BK materials as well as allowing quantitative
 164 comparison between the digital composite designs and the resultant AM physical output. The
 165 image processing workflow was built connecting components in a cascade fashion (Figure 3a).
 166 X-ray tomograms were imported in MeVisLab to define an interactive reconstruction of
 167 Volumes of Interest (VOIs). For the purpose of this study, a visual script aimed at recognising
 168 greyscale distribution in the samples was generated and then followed by a procedure of
 169 segmentation via region growing to discern different phases. Given the similar densities of the
 170 materials used, a filtering procedure based on Gauss smoothing was necessary to homogenise
 171 the greyscale value of the voxels belonging to the same phase. The *GaussSmoothing*
 172 component provided by MeVisLab, performs an isotropic smoothing to the image dataset in

173 input. A gauss sigma equal to 2 was adopted to set the window of the filter kernel. A calibration
 174 step performed on the overall shape of the samples was necessary to verify that the applied
 175 Gauss filtering would maintain the same volume of the original geometry. Threshold values to
 176 segment VOIs were found analysing the histograms of each data-sets or materials combination.
 177 The region growing procedure started with the selection of a *seed* that is a voxel belonging to
 178 the considered XCT image dataset. The region was then expanded to neighbouring voxels in
 179 which grey values are coincident or fall in an interval defined by the user and named Threshold
 180 Interval Size [%]. For this purpose, a seed was placed within the VOI (matrix plus fibre),
 181 adopting a threshold interval size to be decreased in a multistep procedure.

182



183

184 Figure 2. Schematic showing: (a) the overall workflow and the subsequent steps to (b) place the red region
 185 growing pixels cluster (seed) into the core region to be separated from the rest; (c) datasets comparison
 186 before and after Gauss smoothing to homogenise the highlands effects within the fibre core; (d) region
 187 growing procedure starting from the seed previously introduced and (e) the final post-processed imaging
 188 highlighting a discrete fibrous object within a matrix.

189 The multistep procedure aimed to identify the optimum threshold to segment the sample as
 190 multi-material piece. Figure 3b describes how to highlight the VOI decreasing the threshold
 191 interval size. Once the VOI has been segmented, the second step of the region growing
 192 procedure aimed to separate the fibre from the matrix. The two regions are spatially assembled;
 193 therefore, the image dataset was processed applying a Gauss smoothing filtering to homogenise
 194 pixel intensity (Figure 3c). The contrast between fibre and matrix was enhanced to facilitate a
 195 further region growing segmentation, separating fibre from matrix. The fibre itself represents
 196 the new VOI; thus, all the grey values belonging to the matrix were neglected. The starting
 197 seed was placed in the fibre and multistep procedure used to identify the Threshold Interval
 198 Size [%] for both BK fibres into WT matrix and vice versa. The threshold Interval Size [%] in
 199 the multistep was decreased to highlight a core volume comparable to the one of the designed
 200 fibres.

201

202 **Meshes geometric difference**

203 Quantification of geometrical differences between CAD and XCT datasets was performed
 204 using a Hausdorff distance criterion. Specifically, MeshLab v. 1.3.4beta (Visual Computing
 205 Lab, ISTI, CNR) applied a sampling filter called Hausdorff distance, based on the Metro digital
 206 tool³². The tool numerically compares two triangulated meshes representing the same surface
 207 at different levels of detail. In the current study the two surfaces are represented by the ideal
 208 CAD drawing and the resultant reconstructed mesh surface after image processing. The
 209 difference between two meshes results in an approximation error that can be defined as the
 210 distance between corresponding sections of the meshes. Such differences are evaluated by
 211 considering a point p on a surface S , such that a distance to another point can be defined by e
 212 (p,S) thus:

213

$$214 \quad e(p,S) = \min_{p' \in S} d(p,p') \quad \text{Equation (1)}$$

215 Where d is the Euclidean distance between two points in the space E_3 . The distance between
 216 two surfaces S_1 and S_2 is then defined as:

217

$$218 \quad E(S_1,S_2) = \max_{p \in S_1} e(p,s_2) \quad \text{Equation (2)}$$

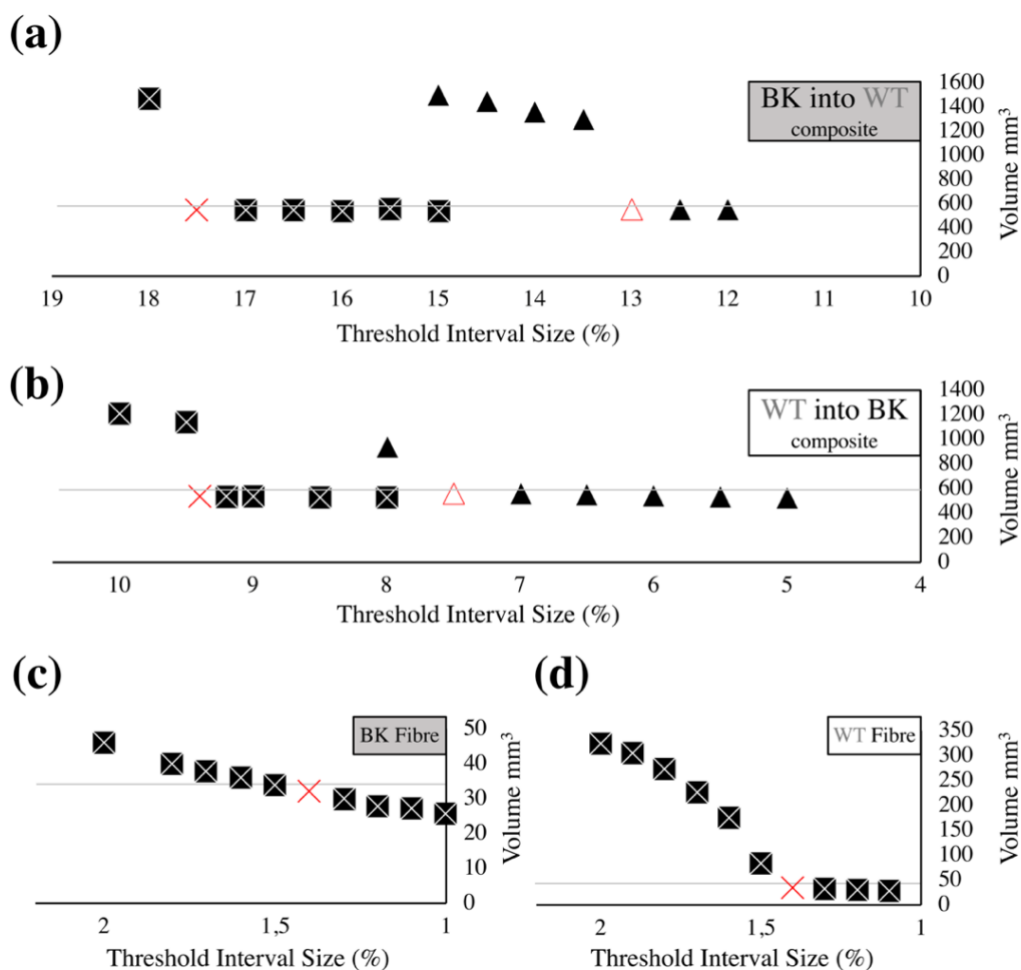
219 This definition of distance is not symmetric such as $E(S_1, S_2) \neq E(S_2, S_1)$. A two-sided distance
 220 (Hausdorff distance) may be obtained by taking the maximum of $E(S_1, S_2)$ and $E(S_2, S_1)$. The
 221 MeshLab command uses a sampling function that computes the aforementioned formula. A
 222 number of points p_1 is defined onto a mesh or surface S_1 . The search is carried on for each point
 223 p_1 to find the closest point p_2 on the mesh or surface S_2 . The software reports numerical results
 224 in mesh units with a red through green to blue map reporting a corresponding increasing
 225 coincidence between the digital design and AM physical object.

226

227 Results

228 Image processing: VOIs thresholding

229 Figure 4 shows a comparison between the volume segmentation before (RAW dataset) and
 230 after the Gauss smoothing.



231

232 Figure 3. Multistep region growing procedure for (a) BK into WT and (b) WT into BK material composites.
 233 The threshold interval size [%] is reported on the x-axis whereas the segmented volume is provided by the

234 y-axis for each iteration step. The black squares represent for the procedure applied to the. RAW datasets.
 235 The triangles draw the trends for the. RAW dataset after Gauss smoothing. The red cross and triangle
 236 represent the best threshold interval sizes for each dataset. Finally, the horizontal line traces the best case for
 237 the segmentation procedure matching the ideal volume of interest from CAD drawings. Trends for best
 238 threshold interval size [%] multistep region growing procedure are depicted for the (c) BK fibre and for the
 239 (d) WT fibre.

240

241 Threshold interval sizes [%] of 17.5% and 13%, around the volume of the original CAD
 242 drawings of 600 mm³, were identified for BK into WT composite RAW data set and Gauss
 243 smoothed (Figure 4a), whereas values of 9.4 % and 7.5 % were found for the WT into BK
 244 composite (Figure 4b). Variation in the recorded volume during each iteration of the growth
 245 are also shown in Figure 4a for BK into WT and Figure 4b WT into BK. The original CAD
 246 drawing defined a target fibre volume of 32 mm³, which was reached in Figure 4c and Figure
 247 4d for optimised threshold interval sizes of 1.2% and 1.4%.

248 Table 1. Geometrical dimensions from CAD drawings compared to those of the reconstructed meshes after
 249 XCT scanning of BK into WT and WT into BK materials. Absolute relative changes are reported as
 250 percentage of the CAD values.

Triangulates surfaces comparison

	CAD	BK into WT	WT into BK	Abs. Relative Difference (%)	
				BK into WT	WT into BK
Matrix Volume (mm³)	600.0	554.2	542.9	7.6	9.5
Fibre Volume (mm³)	32.0	31.4	30.8	1.9	3.7
Matrix Surf. Area (mm²)	669.8	635.6	637.2	5.1	4.9
Fibre Surf. Area (mm²)	72.0	71.2	70.8	1.1	1.7
Matrix width (mm)	10.0	9.9	10.1	1.5	1.1
Fibre width (mm)	2.0	2.6	2.3	30.0	15.0
Matrix length (mm)	24.0	23.9	24.0	0.4	0.2
Fibre lenght (mm)	8.0	8.6	8.1	7.5	1.2
Matrix thickness (mm)	3.0	3.2	3.2	6.6	6.7
Fibre thickness (mm)	2.0	1.9	2.2	5.0	10.0

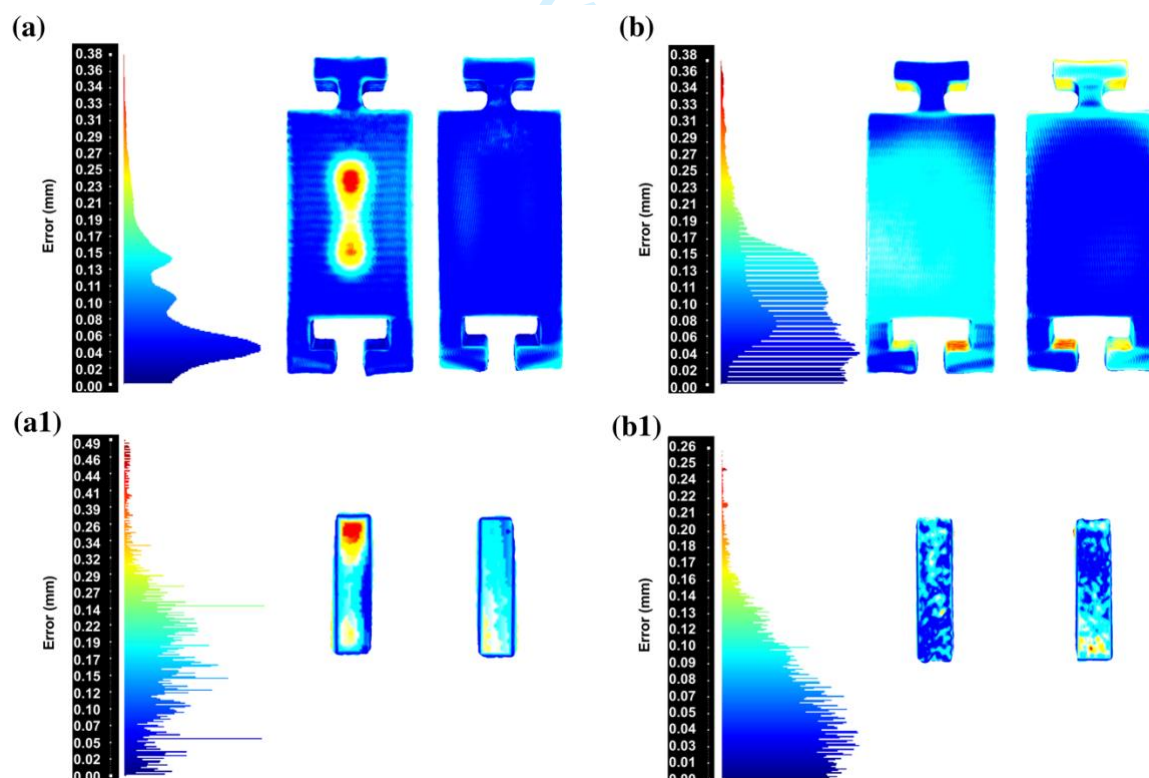
251 A further investigation was carried out in CAD environment to provide insights into the
 252 accuracy of the reconstructed VOIs in terms of linear measurements of width, length and
 253 thickness. Table 1 outlines the results of several geometrical comparisons referring to

254 reconstructed meshes to the original CAD drawing. The analysed parameters were volume,
 255 surface area, width, length and thickness for both the reconstructed fibre and matrix. To have
 256 comparable measurements, geometries were aligned in x, y and z axis and bounding box
 257 volumes retrieved. For consistency, the fibre and matrix lengths, widths and thicknesses
 258 measurements were taken from the bounding box. Surface area and volume were evaluated
 259 again in the 3D modelling environment Rhinoceros.

260 Meshes geometrical differences

261 Resultant Hausdorff distance maps are shown in Figure 5. Red and blue are the boundary
 262 colours representing distances between the meshes (0.00 mm to 0.38 mm) for the matrix and
 263 (0.00 mm to 0.49 mm) for the fibre, respectively. The BK fibre into the WT matrix displayed
 264 maximum Hausdorff distances of 0.38 mm (Figure 5a) and 0.49 mm (Figure 5 a1) for the matrix
 265 and fibre respectively, whereas the WT into BK composite showed smaller distances for matrix
 266 (0.38 mm, Figure. 5b) and fibre (0.38 mm, Figure 5 b1).

267



268

269 Figure 5. Geometrical difference applied to the reconstructed VOIs using the Hausdorff distance in Meshlab
 270 for matrix (top) and fibre (bottom). The reconstructed geometries were compared with the respective CAD
 271 drawings, BK into WT (a, a1) and WT into BK (b, b1). The distance range is visually reported and
 272 represented by a shade of colours that goes from blue to red accompanied by the relative variation.

273 Discussion

274 Multi-material AM avoids multistep parts assembly typical of composites materials
275 manufacturing. A full exploitation of multi-material AM requires a complete examination of
276 the 3D printed builds for modelling purposes and to eventually rely on composite theories for
277 prediction and tuning of the mechanical interplay between composite phases. Although 3D
278 printing can accurately reproduce specific designed models, the prints can deviate from the
279 designed models. For this reason, volumetric information is needed and XCT is clearly
280 advantageous when compared to other imaging techniques that are constrained to 2D
281 investigation. XCT systems have been designed for applications of industrial metrology
282 becoming the only commercially-available, non-destructive method to perform dimensional
283 measurements of internal geometrical features^{33 34}. Among all the added benefits of a non-
284 destructive volumetric investigation, XCT 3D representations of objects can be processed and
285 analyzed to be further compared to their original drawings, showing coincidence between
286 manufactured and digital volumes. So far, XCT has been mainly employed to determine
287 distribution of random inclusion in the same manufactured sample, evaluate distribution of
288 pores, interfacial density variation between different phases as well as degradation of
289 pharmaceutical 3D printed drug delivery dosage forms^{22 20 23 24}. Although material extrusion
290 such as FDM is commonly used in prototyping composite parts, AM technologies such as ink-
291 jet printing, are now capable of producing higher resolution parts with improved densities and
292 mechanical properties³³. Particularly, ink-jetting gained attention as a credible industrial
293 method, due to its scalability and multi-material part production³⁵ and a better understanding
294 of geometrical variations in the 3D printed build relative to the different materials used to
295 produce the multi-material composite is of paramount importance.

296 The aim of this study is to implement a procedure able to qualitatively and quantitatively
297 analyse the direct assembly of ink-jet 3D printed composites. The proposed workflow was able
298 to determine the composition of a single fibre composite that is a simplified version of AM
299 polymeric composites previously proposed^{25 26 27}. Differentiating the core fibre from the bulk
300 within AM composite with XCT resulted challenging mostly due to the presence of phases
301 with similar densities. Most ink-jet print heads work best with inks of low viscosity at/or near
302 room temperature³³. This could be impairing XCT ability to discriminate elements with similar
303 attenuation and dimensions, such as ink-jets. To be effective XCT needs to be coupled with
304 imaging techniques able to reconstruct and partition regions in an image. To this purpose
305 Gaussian smoothing was used to 'blur' images and remove detail and noise. In this sense it is

306 similar to the mean filter but it uses a different kernel that represents the shape of a Gaussian
307 ('bell-shaped') hump. One of the principal reasons for using a Gauss filter is due to its frequency
308 response able to reduce the high spatial frequency components from an image³⁶. Compared to
309 other linear filters such as the Mean filter, gaussian blurring is a linear operator rather than non-
310 linear. Contrary to the mean filters which performs a uniform weighted average, the degree of
311 smoothing of a gaussian filter is determined by the gaussian standard deviation; the output
312 represents a weighted average of each voxel's neighborhood rather than uniform smoothing.
313 Other filters such as the Median are non-linear and are famous to remove details and noise and
314 to preserve edges³⁷. In the case of this study edges were reconstructed combining the Gaussian
315 operator to a region growing procedure due to the fact that edges were not defined. Indeed,
316 what appeared as a cluster of islands in the data sets core region, were joined to form a boundary
317 that reflected the fibre shape. This aspect was fundamental for the XCT visualisation of the
318 fibre core region in this study. Moreover, a complementary materials allocation WT into BK
319 and WT into BK was considered to understand how this could affect the manufacturing process
320 hence deviating from the original CAD design according to different materials allocation. The
321 gauss filter successfully smoothed and interconnect the sub-regions in the fibres, enhancing the
322 contrast between matrix and fibre. Finally, a multistep region growing procedure helped in
323 defining thresholding values to isolate a continuous fibre from the external matrix.

324 Further analysis in terms of surface area and volume were performed on the resulting meshes
325 STL and directly compared to files CAD source after both XCT datasets were converted into
326 STL files to perform a geometrical difference with the CAD drawing. Table 1 allows critical
327 considerations of the investigated material assemblies. The volume of the whole sample was
328 coincident to the CAD model in terms of volume and surface area where the largest discrepancy
329 was shown from the BK fibre into the WT matrix. A decrease in volume for both samples BK
330 into WT and WT into BK of 7.6% and 9.5 % respectively, can be related to either the XCT
331 voxel size or material behaviour during production. The volume of the embedded fibres was
332 almost replicated with the digital design, showing a 3.7% and 1.9% volume reduction in
333 comparison to the original CAD drawing of the fibre. The surface area of the matrix and fibre
334 for both material combinations slightly differed from the design, showing a maximum of 5.1%
335 surface area loss for BK into WT composite. Width, length and thickness of the parts and the
336 fibre all differed for less than a 10% compared to the CAD drawing, except for the widths of
337 the fibre, which were found to be bigger by 30% and 15% for the BK into WT and WT into
338 BK, respectively. This can be explained by referring to the ink-jet AM process, which uses a

339 roller/planerizer in order to flatten the printed parts and remove bubbles and excess of material
340 from the printed layer, according to the resolution set for the AM process³⁸. Critically, the fibre
341 core of the composite, is still partially cured by the UV light when the roller passes onto the
342 layer, spreading the core in the XY plane while pushing the fibre to spread at the interface into
343 the matrix. This appears to be more effective for the combination BK into WT where the black
344 elastomeric material is the fibre.

345 **Conclusion**

346 An original metrology tool applied to multi-material AM composites has been investigated
347 within this study. Two different allocation of materials, BK into WT and WT into BK, were
348 considered to investigate manufacturing fidelity when fibre and matrix were mutually produced
349 with rigid WT and flexible BK materials, but complying with the same CAD drawing. XCT
350 helped to produce a virtual model of the 3D printed part, showing regions very similar in
351 density, hence difficult to be distinguished and compared. In order to stabilize the workflow,
352 the procedure considered a region growing calibration step performed on the overall shape of
353 the samples without considering the embedded fibre. This study demonstrates the viability of
354 XCT to quantitatively analyse AM multi-materials prototypes made of soft and hard phases
355 produced via ink-jet AM. This investigation was relevant in defining design boundary
356 conditions below which features are partially reproduced in respect to the original digital
357 drawings, with the potential of affecting the phases mechanically interplay, which is
358 fundamental for materials science and engineering composite applications.

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