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Influence of waste glass in the foaming process of open cell porous ceramic 1 2 as filtration media for industrial wastewater Andrei Shishkin¹, Hakim Aguedal^{2,3}, Gaurav Goel^{4,5*}, Julite Peculevica¹, Darryl Newport⁶, 3 Jurijs Ozolins¹ 4 ¹Rudolfs Cimdins Riga Biomaterials Innovations and Development Centre of RTU, Institute 5 6 of General Chemical Engineering, Faculty of Materials Science and Applied Chemistry, Riga 7 Technical University, Pulka 3, Riga, LV-1007, Latvia. 8 ²Laboratoire de Valorisation des Matériaux, Département de Génie des Procédés, Faculté des 9 Sciences et de la Technologie, Université Abdelhamid Ibn Badis - Mostaganem, Bp. 227, 10 27000 Mostaganem, Algeria. 11 ³Laboratoire Ressources Naturelles Sahariennes, Département des Sciences de la Nature et de 12 la Vie, Faculté des Sciences et de la Technologie, Université Ahmed Draia – Adrar 01000, 13 Algeria. 14 ⁴School of Engineering, London South Bank University, SE10AA, United Kingdom. ⁵School of Aerospace, Transport and Manufacturing, Cranfield University, MK430AL, UK 15 ⁶Sustainability Research Institute, University of East London, London E162RD, United 16 17 Kingdom. 18 19 *Corresponding author Email address: goelg@lsbu.ac.uk 20 21 Abstract

22 This paper reports the development and testing results of a prototype ceramic filter with excellent 23 sorption properties (< 99% elimination in 5 min) leading to good efficacy in the removal of industrial 24 contaminants (Reactive Bezaktiv Turquoise Blue V-G (BTB) dye). The novelty in the investigation 25 lies in developing the filter material obtained from the recycling of waste glass combined with highly 26 porous open-cell clay material. This newly developed material showed a significant reduction in the 27 energy requirements (sintering temperature required for the production of industrial filters) thus 28 addressing the grand challenge of sustainable and cleaner manufacturing. The methodology entails 29 sintering of the clay foam (CF) at temperatures ranging from 800 to 1050 °C and blending it with 5%, 30 7% and 10 wt.% milled glass cullet. One of the aims of this investigation was to evaluate and analyse 31 the effect of the pH of the solution, contact time and equilibrium isotherm on the sorption process and the mechanical compressive strength, porosity, water uptake. From the kinetic studies, it was 32

discovered that the experimental results were well aligned with the pseudo-second-order model and chemisorption was discovered to be a mechanism driving the adsorption process. These findings are crucial in designing cost-effective industrial filtration system since the filter material being proposed in this work is reusable, recyclable and readily available in abundance. Overall, the pathway for the reuse of waste glass shown by this work help address the sustainability targets set by the UN Charter via SDG 6 and SDG 12.

39 Keywords: Ceramic clay foam, open cell, sorption capacity, textile dye, isotherm.

40 **1. Introduction**

The amount of waste glass, particularly soda-lime glass, is increasing year by year. In the 41 42 European Union, approximately 16.3 million tons of cullet was collected in 2016 (Eurostat, 43 2020). There is a pressing need to recyle waste glass and toaddress the theme of waste to 44 wealth. Accordinly, a number of studies have been conducted to use glass cullet as an 45 additional raw material or as a fluxing agent in ceramics (Ondruška et al., 2019) for instance, 46 as a surrogate to natural aggregates, in concrete to build pavements and roads (Majdinasab 47 and Yuan, 2019), and to obtain stoneware tiles and bricks (Chen et al., 2018; Lu et al., 2019; 48 Walczak et al., 2015). The use of cullet, particularly for municipally recycled glass, panel glass, cathode ray tubes has been explored in greater details. Its usage has also been 49 50 investigated for the manufacturing of glass foams (Silva et al., 2018). Glass foams are 51 produced by combining various foaming agents. Primarily used foaming agents are aluminum nitride, calcium carbonate, silicon carbide and manganese dioxide. The foaming process 52 53 energy requirement depends on the foaming agent: 700-725 °C for CaCO₃ (Bernardo and 54 Albertini, 2006) and 900-1000 °C for SiC (Francis et al., 2013).

55 In an earlier study glass-ceramics or porous components were manufactured using natural materials and industrial waste, and a compression strength of ~ 80 MPa (Binhussain et al., 56 57 2014) was achieved. It has been reported, that the addition of 10 wt.% of waste glass in clay 58 with firing at temperatures of 900-1000°C improved the mechanical traits (Phonphuak et al., 59 2016) and reduced the firing temperature by 80-100 °C (Shishkin et al., 2020). However, very 60 few studies exist concerning addition of glass (Binhussain et al., 2014; Silva et al., 2018). 61 Taking into account clay-mineral diversity, there is extensive gap in the knowledge relating to clay- cullet mix properties. 62

63 With the growing knowledge surrounding the ceramic material with the advantage of 64 their chemical stability, high porousness (Dong et al., 2012), high refractoriness, low mass

and thermal conductivity, as well as specific heat (Fukushima and Colombo, 2012), the 65 interest in these materials is increasing. Work has been done on different aspects such as 66 weight reduction (Goel et al., 2018), improved thermal and acoustic insulation properties 67 (Hostler et al., 2009), application as filters (Song et al., 2006) and catalyst support (Choo et 68 69 al., 2019). During the past decade, a wide assortment of approaches have been investigated 70 for producing absorbent ceramics, including the replica, the sacrificial template, water-oil 71 emulsion and the direct foaming methods (Colombo, 2008; Yue et al., 2011). However, only 72 one solitary study has been devoted to direct foaming technique despite it being fastest and 73 easiest method (Lakshmi et al., 2015).

Wastewater containing residual dye is produced primarily for textile and leather industries 74 75 (Yaseen and Scholz, 2019). The treatment technologies utilised at industrial level are either 76 based on activated carbon or Fenton's reagent based method (Wang et al., 2008). These 77 processes are expensive, particulary the regeneration of activated carbon is a tedious and 78 costly operation. In some cases, further treatments are required to purify the water (Durán-79 Jiménez et al., 2014). So as to overcome the impediments of these technologies, this study 80 proposes porous ceramics as a material for water filtration. Reactive Bezaktiv Turquoise Blue 81 V-G (BTB) dye which is a synthetic anionic dye widely used in the textile industry, was chosen as model organic pollutant. Adsorption of this dye by the designed product was 82 83 studied and related manufacturing conditions such as firing temperature and glass cullet -84 Illitic clay mix ratio were analysed.

85 The overarching aim of this work was to produce clay foam (CF) with high adsorption and adequate compressive strength. The methodology entails sintering of the CF at temperatures 86 87 ranging from 800 to 1050°C and blending it with 5%, 7% and 10 wt.% milled glass cullet. 88 The novelty in investigation lies in utilizing the waste glass combined with open cell porous 89 clay material that showed significant reduction in the sintering temperature (800° C). This 90 research considered various dye mixed water properties including pH (2-11), contact time (5-300 min), and dye concentration (10-900 mg \cdot L⁻¹). Optimum combinations of these factors 91 92 were determined to maximize the dye adsorption by clay foams, and the mechanisms involved 93 in the process were examined.

The rest of the paper is organised as follows: Section 2 provide details on fabrication technique for clay foam along with batch experiments for dye adsorption. Section 3 details the experimental results and relevant discussions. The paper concludes by highlighting that sorption kinetic process follows the pseudo-second order kinetic model and chemisorption was the dominant mechanism.

99 **2. Materials and methods**

Details of materials, fabrication methods, chemical and physical characterization techniques,and batch adsorption experiment details are discussed below.

102

103 2.1. Reagents and Raw Materials

104 The illite type clay (deposition Liepa, Lode JSC, Latvia) was used for foam preparation. 105 This homogenised clay was secured from Lode Ltd brick factory (Latvia). The clay agglomerate were dried at 105°C for 24 h, followed by refining in a jaw crusher to yield 106 107 particle sizes in the range of 10-20 mm. It was then subjected to milling using a laboratory 108 disintegrator DSL-175 (designed at Tallinn Technical University (Estonia) (Zimakov et al., 109 2007))for obtaining particle sizes $< 50 \mu m$. Green bottle glass was used as an milled cullet 110 (MC) source. Clean bottles were fed in jaw crusher, in order to obtain particle size of 10-20 111 mm. Subsequently they were also milled using the DSL-175 to yield particle sizes $< 50 \,\mu$ m. 112 Detailed analysis of chemical and mineralogical composition of the clay and cullet used here 113 can be referenced from our previous work (Shishkin et al., 2020) but some salient details are 114 provided here. Oxide composition of the used clay are: $SiO_2 - 62.8$ wt%, $Al_2O_3 - 15.4$ wt%, 115 Fe₂O₃ - 6.8 wt%, CaO - 0.7 wt%, MgO - 1.4 wt%, Na₂O - 0.1 wt%, K₂O - 4.2 wt% TiO₂ -116 1.9 wt%; And for MC SiO₂ – 70.2 wt%, $Al_2O_3 - 2.1$ wt%, $Fe_2O_3 - 0.1$ wt%, CaO - 9.5 wt%, 117 Na₂O - 16.6 wt%, Fatty alcohol sulfate preparation - Schäumungsmittel W53 Flüssig 118 (Zschimmer & Schwarz GmbH & Co KG Chemische Fabriken, Germany) was used as 119 foaming agent (FA). Municipal utility tap water (Riga, Latvia) and deflocculation agent made 120 of humates and silicates basis Dolaflux B11 (Zschimmer & Schwarz GmbH & Co KG 121 Chemische Fabriken, Germany) were also used for the clay slurry preparation. Phase composition after adding glass was as follows: (i) 800 °C- Quartz, Illite, Haematite, 122 123 Microcline, Diopside, and amorphous phase (ii) 900 °C- Quartz, Illite, Haematite, Microcline, 124 Diopside, Spinel, Cristobalite, and amorphous phase.

125

126 2.2. Mixing and milling

127 The dry clay and cullet was milled and homogenised using the laboratory disintegrator 128 DSL-175 at a speed of 12000 rpm. Clay foam (CF) was obtained by direct foaming using 129 similar process as reported in previous study (Shishkin et al., 2015) utilising high-speed 130 mixer-disperser device (HSMD) (Corvus Ltd., Riga, Latvia). The core idea of the HSMD is 131 focused on ceramic slurry treatment producing multiple impacts in a liquid medium

132 (cavitation effect) by dispersing elements through the intensive homegenisation thus
133 introducing micro (50-100µm) bubbles (direct foaming) in bulk product.

134

135 2.3. CF Production Method

136 The mix of clay with MC content of 5%, 7%, and 10 wt.% was prepared as the first step. 137 Preparation of the CF comprised of the following stages. The HSMD was operated in 138 circulation mode and set at 500 rpm. It was filled with 600 ml of water and 1 wt.% of 139 dispersant (calculated from dry clay-glass mixture – 14 g). A total of 1400 g dry clay-MC mix was slowly (~300 g.min⁻¹) added to prevent agglomeration and the HSMD speed was 140 gradually increased to 4000 rpm. The 5.5 wt.% of foaming agent (calculated from dry clay-141 142 glass mixture -77 g) was added after 30s and the mixer speed was gradually raised to 6000 143 rpm. With air introduction into the suspension, the foamed mix volume increased twofold. Treatment by HSMD was continued in recirculation mode (1 min). The CF was then fed into 144 145 a mould of size 150x150x60 mm and naturally dried for 72 hours at room temperature. The 146 sample was additionally dried at 105°C for 24h. Oven dried sample were removed from the 147 mould, cut into 55x55x110 mm specimens and fired in a muffle furnace (LH11, P330 by Nabertherm) at 900, 950, 1000 and 1050 °C at a rate of 5 °C min⁻¹ with a dwell time 30 min. 148 149 The sintered CF samples were cut and polished to a final size of 50x50x50 mm for further 150 tests.

151

152 2.4. Determination of the Physical and Mechanical Properties

153 Firing shrinkage was evaluated by direct sizing of the sample's linear dimensions 154 (55x55x110 mm) before and after firing. The water absorption, apparent density, bulk density 155 and apparent porosity were analysed by the pycnometry and Archimedes method (Annual 156 book of ASTM standards, ASTM Standards C20, 2015). Compressive strength of the sintered 157 specimens (25x25x25mm) were assessed using a Universal Testing Machine (UTM) (Instron 158 8801, Germany) in accordance with ASTM D695. All experiments were organized based on a 159 set of six samples for each category (glass addition and firing temperature) and results 160 reported.

For the microstructural characterization, optical microscope (Keyence Corporation, Osaka,
Japan, VHX-2000 and VH-Z20R/W lens) was used. Scanning electron microscope (SEM)
Zeiss EVO MA-15 (Carl Zeiss AG, Oberkochen, Germany) was used for microstructural
characterization at high magnification. High temperature optical microscope (EM201 HT163,
Hesse instruments, Germany) was used for optical dilatometry.

166

167 2.5. Preparation of dye solution

A synthetic dye (BTB), widely used in the textiles industry, was chosen as the targeted pollutant to gauge the sorption capacity of the fabricated specimens and their probable application in industrial wastewater purification. Stock solution was formed by dissolving exactly weighed amounts (1g) of dye in bi-distilled water to give a strength of 1000 mg \cdot L⁻¹. From this solution, other solutions were prepared by successive dilutions when necessary.

173

174 2.6. Batch adsorption experiments

The adsorption experiment was performed by mixing 25 ml of dye solution and 0.65 g of adsorbent. The mixture was shaken at 200 rpm and placed in a centrifuge for 5 minutes at 4000 rpm. The residual concentration of dye was determined by analysis of the supernatant at the wavelength that is consistent to the extreme absorbance, at 625 nm, using UV-visible spectrophotometer (Evolution 300, Thermo Scientific). Portion of adsorbed BTB was as per subsequent equation (1):

$$q_e(\mathrm{mg} \cdot \mathrm{g}^{-1}) = \frac{(\mathrm{C}_{\mathrm{i}} - \mathrm{C}_e) \times \mathrm{V}}{\mathrm{m}}$$
(1)

181 where q_e is the quantity of dye adsorbed by bulk of adsorbent (mg·g⁻¹), C_i and C_e are 182 respectively the initial and residual concentration of dye solution (mg·L⁻¹), m is the amount of 183 adsorbent (g), and V is volume of solution (l).

184

185 The effect of various factors on the adsorption capacity were examined:

- The outcome of pH on the adsorption capacity, was evaluated by using various pH solutions ranging from 2 to 11. The solutions pH were adjusted by using diluted mix of HNO₃ and NaOH (0.1 N).
- Impact of contact time was studied from 5 to 300 min at three altered initial concentrations 50, 100 and 200 mg·L⁻¹. The obtained kinetic results were modelled by Elovich, Pseudo 1st order, Pseudo 2nd order and intraparticle diffusion kinetic models.
- The effect of initial concentration on dye adsorption capacity was investigated at
 20 °C with different dye mixes ranging from 10 to 1000 mg·L⁻¹. Freundlich, Langmuir
 and Temkin adsorption isotherm were utilized to represent the obtained data from
 adsorption isotherm study.
- 196
- 197 **3. Results and discussion**

198 Results on clay foam are presented below for their physical, mechanical properties, and199 adsorption performance.

200 *3.1. Samples and pore morphology*

201 The sample of the CF (MC-7 wt.%), shown in Fig. 1a, had no visible fractures or cracks 202 compared to the CF without MC addition (provided in supplementary information). The 203 microstructure demonstrates (Fig. 2b) an even pore distribution in the sample volume and the 204 presence of interconnected pores which points to an open cell material classification. The 205 sintering process slightly affected the pore size (due to shrinkage), but not the foam 206 morphology where a spherical-like and interconnected open cell microformation was noted 207 for all 12 samples series. The average cell diameter for the sintered CF was in the range of 50-208 200µm and the cell window measurement was 30-70µm (Fig. 1b and 2a). The pore 209 interconnection was formed due to the thin suspension films between the bubbles rupturing 210 during dehydration since the dislocation of powder from the boundary is thermodynamically 211 inimical. The CF walls have a porous structure (Fig. 2b) due to the natural residual clay 212 porosity caused by water evaporation and by formation of lamellar clay structure. The porous 213 walls facilitate the liquid permeability through the structure. The MC addition did not play 214 significant role on the foam morphology and average cell sizes, which were in the 50-200µm 215 range for all samples (5, 7 and 10 wt.% of MC addition).



216 217

218

Fig. 1. CF with 7 wt.% MC content. a) green sample after drying; b) fired at 900°C, optical microscope at magnification X200 times.

- According to micro optical observations, with an escalation in the firing temperature the
- pore sizes decrease by 7-10% for the series sintered at 1050°C in comparison to 900°C. This
- could be explained by the natural clay shrinkage and densification during the firing.



- Fig. 2. SEM micrographs of the CF with 7 wt.% MC content, fired at 900°C
- 225 226 227
- *3.2. Physical properties*

The compressive strength and total porosity of CF samples with diverse MC ratio are 229 230 presented in Fig. 3. It can be perceived that the porosity augmented by 2-3% with an MC 231 increase from 5 to 7 wt.% (for all sintering temperatures). At the same time, the porosity 232 decreases by 2-2.5% with a firing temperature increase by 50°C for each series. The porosity 233 dependence on the firing temperature exhibited a linear tendency. However, for the firing 234 temperature increase from 950 to 1050 °C, the bulk density showed a steady increase for all compositions, from 0.80 to 0.90 g·cm⁻³, from 0.73 to 0.84 g·cm⁻³, and from 0.66 to 0.75 g·cm⁻¹ 235 ³ for 5, 7 and 10 wt.% MC content, respectively. For porous ceramics the bulk density has a 236 237 direct correlation with shrinkage, the higher the sintering temperature the denser the ceramic 238 particles pack and at the same time pore merging can take place.

Compressive strength increases for all series, with the firing temperature increase. However, the intensity of increase is highly dependent on the MC content. At 900°C clay particles could be bonded and in this case, the amount of MC plays a more significant role. The result of 7 wt.% MC composition shows a higher value of compression strength at 900°C. Overall, the low mechanical properties of 10 wt.% MC sample could be explained by the lower mechanical properties of the glass phase. The total porosity varies almost linearly with sintering temperature increase for all compositions (Fig. 3).



Fig. 3. Dependence of the compressive strength on the firing temperature of samples with 5,7
and 10 wt.% of MC loading at different firing temperatures. MC ratio is indicated in the
legend. The firing temperature is indicated in °C.

251 In view of definite mechanical properties - specific strength (Φ/Δ) is of vital significance in 252 lightweight product design and determination of the best material (Ashby, 2011). From the 253 engineering point of view, lightweight panels could be described and evaluated by specific strength Φ/Δ (MPa·g⁻¹·cm⁻³) (Novais et al., 2015; Ponsot et al., 2015). However, from an 254 255 energy efficiency viewpoint it is necessary to consider one more aspect – the temperature and time required for the material production. In the case of investigated natural clay foam with 256 the addition of 5 wt.%, 7 wt.% and 10 wt.% of ground MC, optimal parameters were found 257 258 for 7 wt.% MC. This composition has the highest compressive strength in the 900°C series -259 5.2 MPa. The composition with 5 wt.% MC, sintered at 1000°C has the highest compressive 260 strength in the 1000°C series – 12.9 MPa, and the second highest value of compressive 261 strength (12.6% less than the highest value - 14.3 MPa) in all the series. Therefore 262 composition with 5 wt.% MC, sintered at 1000°C, is better for the construction application (with higher mechanical properties). The record of absolute (MPa) and specific (MPa·g⁻¹·cm⁻ 263 264 ³) strength of the 5-10 wt.% MC loaded clay foam are provided in Table 1.

266	Table1.	The	absolute	Φ	(MPa)	and	specific	Φ/Δ	(MPa·g ⁻	¹ ·cm ⁻³)	strengths	of	MC-clay
267	foam.												

Einin a	MC content, wt.%							
Firing -	5 wt.%		7 wt.9	10 wt	10 wt.%			
temperature -	Φ	Φ/Δ	Φ	Φ/Δ	Φ	Φ/Δ		
800 °C	1.3±0.2	1.9	1.5 ± 0.1	2.3	2.0 ± 0.2	3.6		
850 °C	2.0±0.2	2.9	3.8±0.2	5.8	3.0±0.3	5.2		
900 °C	3.8±0.3	5.3	5.2±0.3	7.7	3.3±0.7	5.3		
950 °C	8.9±0.4	11.1	5.8±0.6	7.9	3.5±0.5	5.3		
1000 °C	12.9±0.1	15.3	9.5±0.1	12.0	4.6±0.3	6.5		
1050 °C	14.3±0.5	15.8	11.1±0.6	13.2	6.3±0.9	8.4		

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270

3.3. Effect of firing and MC content on the adsorption capacity

Preliminary adsorption tests were conducted to evaluate the outcome of firing temperatures (800, 900, 950, and 1000 °C) and glass contents (5, 7 and 10 wt.%).

As demonstrated in Fig. 4, the highest adsorption capacity was observed for samples prepared at relatively low temperatures (800, 900°C) containing 10 wt.% of glass content. The uttermost adsorption capacity registered was 1.7 and 1.8 mg.g⁻¹ respectively for MC10-800 and MC10-900. The lowest adsorption capacity was observed at high firing temperatures. Therefore, compositions MC10-800 and MC10-900 were selected as adsorbents for further evaluation of the adsorption process of BTB dye.

279



281

Fig. 4. Preliminary adsorption test of different prepared materials.

282 *3.4. Influence of pH*

The pH of mix significantly influences the adsorption capacity of dye onto MC10-800 and MC10-900 as shown in Fig. 5. Indeed, the adsorption capacity augmented when the pH of solution reduced, with the maximum adsorption capacity for both materials being observed at pH 2. Adsorption capacity of 2.4 and 1.7 mg \cdot g⁻¹ was obtained, respectively, for MC10-800 and MC10-900. These capacities decreased when the pH of solution increased, achieving the lowest values at pH 5.

289 As explained by earlier study (Aguedal et al., 2018), the high amount of BTB dye 290 adsorbed onto raw and thermally treated diatomite, in the acidic range, is due to the 291 protonation influence of superficial functional groups of the adsorbents. This effect generates 292 a high electrostatic attraction among the positively charged adsorbent surface and the dye that 293 seems anionic character. However, in the alkaline pH range, hydroxyl ions compete with the 294 molecules of dye to occupy the same available active adsorption sites, which translates into 295 the lowest amount of dye being adsorbed in this range. Hence, a pH 2 is used in the kinetic 296 and isotherm adsorption studies.

297





300



The rate at which the adsorbent is able to uptake the ascorbate is one of the most significant parameters that governs the appropriateness for the purpose of water quality control. The kinetic study was done to analyse the mechanism and determine the efficiency of the adsorption process. Three concentrations of dye (50, 100 and 200 mg·L⁻¹) were utilized to analyse the adsorption kinetic of BTB dye onto MC10-800 and MC10-900 at different time intervals, between 5 and 300 min, and at an optimised pH. The results are depicted in Fig. 6 and Fig. 7.

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Fig. 6. Sorption kinetics for MC10-800.



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Fig. 7. Sorption kinetics for MC10-900.

314 It was observed, for all selected dye concentrations, that the adsorption capacity of BTB 315 dye augmented with, surge in contact time and then the adsorption rate reduced and tends 316 towards equilibrium. In addition, the surge in the initial dye concentration increased the 317 exclusion efficiency of dye and significantly increased the equilibrium time. At lower 318 concentrations, the number of available adsorption sites is higher than the number of dye 319 molecules in solution, which lead to equilibrium rapidly. However, at higher concentrations, 320 the amount of dye molecules experiences high gradient of concentration at the liquid-solid 321 interface which increase the deep molecules diffusion to the inner part of MC10-800 and 322 MC10-900 adsorbents, thus explaining the rise in adsorption capacity of dye and equilibrium 323 time (El Haddad et al., 2014). The same observations have been reported in other work (Iddou 324 et al., 2011; Novais et al., 2018).

Adsorption kinetic behaviour of BTB dye onto MC10-800 and MC10-900 was studied using the experimental kinetic specifics and analysed by the Elovitch, pseudo-first order, pseudo-second order and intraparticle diffusion kinetic models (see Table 2). The results of the correlation coefficient and the calculated constant bounds for each model are summarised in Table 3.

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Kinetic models	Equations	Linear expression	Plot	References		
Elovich	$q_t = \beta_E \times ln \; (\alpha_E \times \beta_E \times t)$	$q_{t} = \beta \times \ln (\alpha \times \beta) + \beta \times \ln (t)$	q _t vs. ln t	(Zhang and Stanforth, 2005)		
Pseudo 1 st order	$q_t = q_e [1 - exp (-k_{1p} \times t)]$	$\ln (q_e - q_t) = \ln (q_e) - k_{1p} \times t$	$\ln(q_e - q_t)$ vs. t	(Aravindha n et al., 2007)		
Pseudo 2 nd order	$q_t = k_{2p} \times Qe^2 \times t/(1 + k_{2p} \times Q_e \times t)$	$t/q_t = (1/k_{2p} \times q_e^2) + (t/q_e)$	t/q _t vs. t	(Ho and McKay, 1999)		
Intra-particle diffusion	$q_t = k_p \times t^{0.5}$	$q_t = k_p \times t^{0.5}$	$q_t vs. t^{0.5}$	(Arami et al., 2008)		

 q_t and q_e are respectively the adsorbed quantity of dye at time t, and at equilibrium $(mg \cdot g^{-1})$. k_{1p} , k_{2p} and k_p are respectively the kinetic model constant of pseudo-first order (min^{-1}) , pseudo-second order $(g \cdot mg^{-1} \cdot min^{-1})$ and intraparticle diffusion $(g \cdot mg^{-1} \cdot min^{-0.5})$; α $(mg \cdot g^{-1} \cdot min^{-1})$ and β $(g \cdot mg^{-1})$ are the Elovich kinetic model constants.

From the high values of correlation coefficient R^2 , the adsorption kinetics of BTB dye are well matched by the pseudo-second order kinetic model for all adsorbents. In addition, the calculated adsorption capacities by this model are similar to those obtained in batch mode test (see Fig. 8 and Fig. 9).







Fig. 9. The adsorption amount of dye onto MC10-900 obtained experimentally and fitted
 pseudo-second order kinetic model.

Matching of experimental results to the pseudo-second order kinetic model indicates that 18 19 the chemisorption could be the rate-controlling step on the first stage as well as at low 20 concentrations. However, the assumption of molecular diffusion could be one more 21 mechanism involved in the removal process of BTB dye onto MC10-800 and MC10-900, 22 which is justified using intraparticle diffusion kinetic model. Indeed, as reported by Weber 23 and Morris (W. Weber, J. Morris, 1963), if intraparticle diffusion is the dominant step in the adsorption method of BTB dye onto MC10-800 and MC10-900, then a plot of qt versus t^{0,5} 24 25 ought be linear and advance along the origin (Behnamfard and Salarirad, 2009). From Fig. 26 10a and Fig. 10b it can be seen that the plot is often multilinear (two linear steps) signifying 27 that the intraparticle diffusion is by no means the rate-limiting step for the entire adsorption process. The initial linear step confirms the fast adsorption during the first time of contact. As 28 29 the reaction constant for the intraparticle diffusion k_p increase with the initial dye 30 concentration, the slowdown in the second stage is certainly owing to the resistance diffusion 31 of dye molecules into the interior adsorption available sites (Bhatti and Nausheen, 2014). 32





Fig. 10 Intraparticle diffusion kinetic model plots of BTB dye adsorption at different initial concentration onto (a) MC10-800, (b) MC10-900.

Table 3. Kinetic model parameters for MC10-800 and MC10-900 samples.

Kinetics		1	MC10-800		MC10-900		
i inches	Parameters	50	100	200	50	100	200
model		$mg \cdot L^{-1}$					
	β	0.288	0.61	1.136	0.14	0.28	0.66
Elovich	α	3.76	1.94	0.37	9.67	1.97	0.52
	R^2	0.996	0.991	0.953	0.98	0.93	0.86
Pseudo-first	q _e	1.28	2.11	4.961	0.53	1.75	3.53
order	k _{1p}	0.017	0.013	0.012	0.012	0.02	0.012
order	\mathbb{R}^2	0.945	0.928	0.96	0.94	0.89	0.98
Pseudo-	q _e	1.67	3.42	6.38	0.88	1.68	4.1
second order	k _{2p}	0.036	0.015	0.004	0.067	0.017	0.003
second order	\mathbb{R}^2	0.995	0.995	0.983	0.99	0.97	0.91
Interparticle	k _p	0.038	0.177	0.352	0.041	0.09	0.22
diffusion	R^2	0.861	0.871	0.954	0.85	0.96	0.98

44 *3.6. Adsorption isotherm*

45 The adsorption equilibrium isotherm was assessed by studying the outcome of initial 46 dye concentration at room temperature and optimized value of pH and contact time (pH = 47 2, equilibrium time of 2h). Tests were performed at varying dye concentrations from 10 to 900 mg·L⁻¹. Fig. 11a and Fig.11b depict, that the rise in the early dye concentration causes 48 49 a rapid growth in the amount of dye adsorbed onto MC10-800 and MC10-900, and 50 gradually increases until the equilibrium and/or saturation state of MC10-800 and MC10-51 900 adsorbent is reached. The same phenomenon was observed by Kousha et al (2012). As reported by Novais et al (2018), that the mass transfer resistances at the solid-liquid 52 53 interface is overcome by increasing the preliminary dye concentration, which provide 54 significant driving force enhancing the adsorption process. Hence, a higher initial 55 concentration could lead to adsorbent saturation.







Fig. 11. Effect of initial dye concentration (a) MC10-800 and (b) MC10-900.

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Equilibrium isotherm equations characterize the adsorption mechanism of BTB dye onto MC10-800 and MC10-900 adsorbents. Observed data were fitted by Langmuir, Freundlich and Temkin isotherm model (see Table 4). In order to decide if the isotherm model is suitable to describe the experimental results, high correlation coefficients (R^2) for the best fitting isotherm model can be referred. Table 5 summarises all the correlation coefficient R^2 values obtained from each of the models applied in the adsorption study of BTB dye onto MC10-800 and MC10-900.

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Isotherm model	Equations	Linear expression	Plot	References
Freundlich	$q_e = K_F \times \left(C_e\right)^{1/n}$	$ln (q_e) = Ln (K_F) + (ln (C_e)/n)$	ln (qe) vs. ln (Ce)	(Chan et al., 2012)
Langmuir	$q_e = (K_L \times Q_m \times C_e) / (1 + K_L \times C_e)$	$1/q_e = (1/K_L \times q_m \times C_e) + (1/q_m)$	$1/q_e$ vs. $1/C_e$	(Secula et al., 2011)
Temkin	$q_e = q_m \times ln (K_T \times C_e)$	$q_e = q_m \times ln (K_T) + q_m \times ln (C_e)$	q _e vs. ln (C _e)	(Behnamfard and Salarirad, 2009)

Table 4: List of non-linear equations of the used isotherm models and their linearized expressions.

 $q_e (mg \cdot g^{-1})$ and $C_e (mg \cdot L^{-1})$ are respectively the equilibrium adsorbed quantity and equilibrium concentration, $K_F (mg \cdot g^{-1})$ Freundlich isotherm constant $(mg \cdot g^{-1})$, 1/n Freundlich exponent, $q_m (mg \cdot g^{-1})$ maximal adsorption quantity, $K_L (L \cdot mg^{-1})$ Langmuir isotherm constant, $K_T (L \cdot mg^{-1})$ Temkin isotherm constant.

Isotherm model	Parameters	MC10-800	MC10-900
	K _F	0.28	0.19
Freundlich	n	1.19	1.82
	R^2	0.87	0.76
	K _L	0.023	0.018
Langmuir	$q_{\rm m}$	8.80	3.16
	\mathbb{R}^2	0.97	0.96
	K _T	0.28	9.98
Temkin	$q_{\rm m}$	4.36	1.32
	R^2	0.60	0.75

Table 5. The isotherm model paran	neters for MC10-800 and MC10-900
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From Table 5, it may be perceived that the adsorption isotherm data of BTB dye onto MC10-800 and MC10-900 are best matched by the Langmuir isotherm model giving the highest R^2 comparing to Freundlich and Temkin models. Underpinning the Langmuir assumptions, the adsorption molecules of BTB dye onto MC10-800 and MC10-900 occurred in monolayer at a homogeneous adsorbent surface. Thus chemisorption is the main mechanism controlling the whole adsorption process (Caner et al., 2009).

The increasing sorption ability of the MC10-800 material can be explained by the high porosity of CF due to a lower firing temperature. This is supported by the high temperature microscopy results (Fig. 12a and 12b), where it is clearly visible that at temperature above 860°C, the glass changed from solid conditions, corresponding to low-viscosity state and transfers into a semi-liquid phase. At 900 °C glass has much lower viscosity. Beyond 900 °C glass addition promotes Liepa clay sintering, which decreases porosity and increases shrinkage which in good correlation with obtained strength-porosity results for studied porous ceramics in this research.

This is supported by the dilatometry measurements which demonstrated a rapid increase of the shrinkage (Fig. 12a) after 860°C. At the same time, at 860°C, the glass completely transferred to semi-liquid conditions (Fig. 12b) and started working as a ceramic particle binder. Thereby, conditions are optimum at 800°C for MC utilisation with maximum porosity and results in maximum adsorption in the specimen.



Fig. 12. (a) Dilatometry measurements of the MC5, MC7, MC10 (b) high-temperature microscopy measurements of the glass prism cross-section during firing.

3.7. Benchmarking of results and possible mechanism of dye adsorption

Removal of BTB with other adsorbents used by other researchers has been compared in Table 6. Values reported in the table implies that adsorption capacity achieved in this study is comparable with other studies in literature. However it may be noticed while others materials

(e.g. activated carbon) are carbon based and may be for single use whereas ceramic material obtained in this work can be reused for several cycles after thermal treatment or it can be used as construction material (Petrella et al., 2018). Thus, it is cost efficient and environment friendly.

Adsorbent	Adsorption capacity (mg.g ⁻¹)	Reference		
MC10-800	26.8	Present study		
Diatomite ^a	27.8	(Aguedal et al., 2018)		
Activated sludge	38.2	(Djafer et al., 2016)		
Chitosan/zeolite A	305.8	(Nešic et al., 2013)		
Brown macroalga	35.6	(Kousha et al., 2012)		
Pine Cone	37.4	(Mahmoodi et al., 2011)		
Humin immobilized on silica	19.4	(Jesus et al., 2011)		
Activated carbons ^b	66.8	(Ahmad and Rahman, 2011)		
Anarchic sludge	117.1	(Caner et al., 2009)		
Allaeloble sludge	100.0	(Kargi and Ozmihci, 2005)		
Activated carbons ^c	3.4	(Amin, 2008)		
Chitosan-crosslinked beads	30.0	(Kimura et al., 2002)		

Table 6. Removal of the BTB dye with different adsorbents as reported in literature.

^a: Treated at 600°C

^b : Prepared from coffee husk

^c : Prepared from bagasse pith

Based on literature and according to the effect of pH on the adsorption process, possible scheme of the BTB dye adsorption mechanism on the fired clay-glass surface is as follows. All the reactive dyes behaves similarly in aqueous media (Awasthi et al., 2020). As noticed from molecular structure of BTB dye (Fig. 13), it will be ionized in aqueous solution, by the presence of sulfonate group (SO³⁻) groups and the rupture of the OH bond leading to an overall negative charged aqueous solution. On the other hand, in acidic range, the protonation effect conduct to charge positively all the functional groups on the surface of ceramic clay (Al-Ghouti et al., 2009). Hence, a strong electrostatic attraction is created between the dye molecules and the surface of ceramic clay according to the scheme (Fig. 14). Once, bonds structure is occupied, further links could occur between the dye molecules via intermolecular attraction such as Van der Waals and hydrogen bonds (Ahmed and Jhung, 2017). Additionally, the O–H or oxygen bonds present on CF (Yang et al., 2020) may probably form an n– π bonding with the aromatic ring of BTB dye (Olusegun and Mohallem, 2020). The decrease of the adsorption capacity at high pH is due to, at alkaline range, the hydroxyl ions will be adsorbed strongly on the active site onto ceramic clay competing thereby with the

BTB dye molecules, hence leading to the high repulsive interactions for the BTB dye molecules (Awasthi et al., 2020).



Fig. 13. BTB dye chemical structure (ChemicalBook, 2020).



Fig. 14. Scheme of the possible adsorption process mechanism.

Conclusion

This novel multidisciplinary research paper reports experimental aspects involved in the development of new material (potentially useful as a component of industrial water filtration system). The material fabricated is highly porous clay ceramic foam containing cullet and is

obtained by direct foaming and cavitation disperser. Due to the relatively simple technology of ceramic clay foam production, with careful technological development, the technological uptake of the method is sustainable and it addresses the pressing international need of bringing environmental-friendliness in industrial production. Future development of the clay-glass ceramics should be directed towards lowering of firing temperature below 800 °C and improved glass milling efficiency to obtain finer particles with higher specific surface area to achieve greater sorption ability.

During this investigation, it was discovered that the mechanical properties improve with the increasing glass content and perform best when the fired temperature is in the range of 800-900°C. Other broad conclusions drawn were as follows:

- Benchmarking done against the published literature showed the newly proposed material possess a desirable adsorption capacity of 26.8 mg.g⁻¹ comparable to other research materials developed and reported to date. Furthermore, the open cell ceramic foam promotes a cleaner alternative.
- Experimental results indicated that the glass transferred to liquid phase above 860°C and started promoting intensive bonding with ceramic particles. This bonding contributes towards higher compressive strength at higher sintering temperatures.
- A pH value of 2 was found most optimal to attain the maximum adsorption capacities of up to 13%. The equilibrium time was affected by the intensification in the initial dye concentration.
- The sorption kinetic process followed the pseudo-second order kinetic model, indicating potential chemical relations between the sorbent and the sorbate. The Langmuir isotherm model described the experimental results well compared to other empirical models. Thus, chemisorption was discovered to be the driving mechanism controlling the adsorption process.

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Highlights

- Direct foaming method for formulating an open cell ceramic material.
- Maximum sorption capacity of 26.8 mg \cdot g⁻¹ for sample fired at 800 °C.
- Cullet transferred from glassy to liquid conditions and promoted intensive bonding.
- Chemisorption is the main mechanism controlling the whole adsorption process.

Journal Pre-proof

Andrei Shishkin: Conceptualization, Methodology, Experimentation,
Writing - original draft, Writing - review & editing. Hakim
Aguedal: Conceptualization, Methodology, Formal analysis, Writing - original draft, Writing - review & editing. Gaurav
Goel: Conceptualization, Writing - original draft, Writing - review & editing. Julite Peculevica: Methodology, Experimentation,
Investigation. Darryl Newport: Methodology, Writing - review & editing.
Jurijs Ozolins: Project administration, Data curation.

Journal Pre-proof

Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Journal Pre-proof